

Pseudopotentials for interacting atoms

Extended capabilities of pseudopotential approach based on
all-electron pseudopotentials

Contents

- Standard pseudopotentials
 - norm-conserving pseudopotentials
 - ultrasoft pseudopotentials
- Desirable pseudopotential properties
 - “softness”
 - “transferability”
- All-electron pseudopotentials
 - construction process
 - properties, comparison to standard pseudopotentials

What is pseudopotential:

operator simulating the effect (within LDA):

of [NUCLEUS + CORE ELECTRONS]

on electronic states

in the energy range of interest, e.g. valence states,
unoccupied states (LDA!) (i.e. not on any state !!!)

in the real-space region significant for chemical bonds

Requirements, expected properties:

- “sufficient” accuracy in wide energy range (transferability)
- real merit to computational efficiency and/or accuracy
 - by reducing the basis set size (compared to AE)
 - by eliminating large energies of core states

Standard pseudopotentials - history

- *Empirical pseudopotentials*
parameters tuned so that PSP give “good results” for some particular structure; then PSP is assumed to be transferable to another structure
- *Ab-initio pseudopotentials*
 1. free atom calculation; for the atomic state of interest
 2. construct pseudowavefunction for a given l and E

$$\psi_{l,E}^{\text{PS}}(r) \equiv \psi_{l,E}^{\text{AE}}(r), \quad r > R_C$$

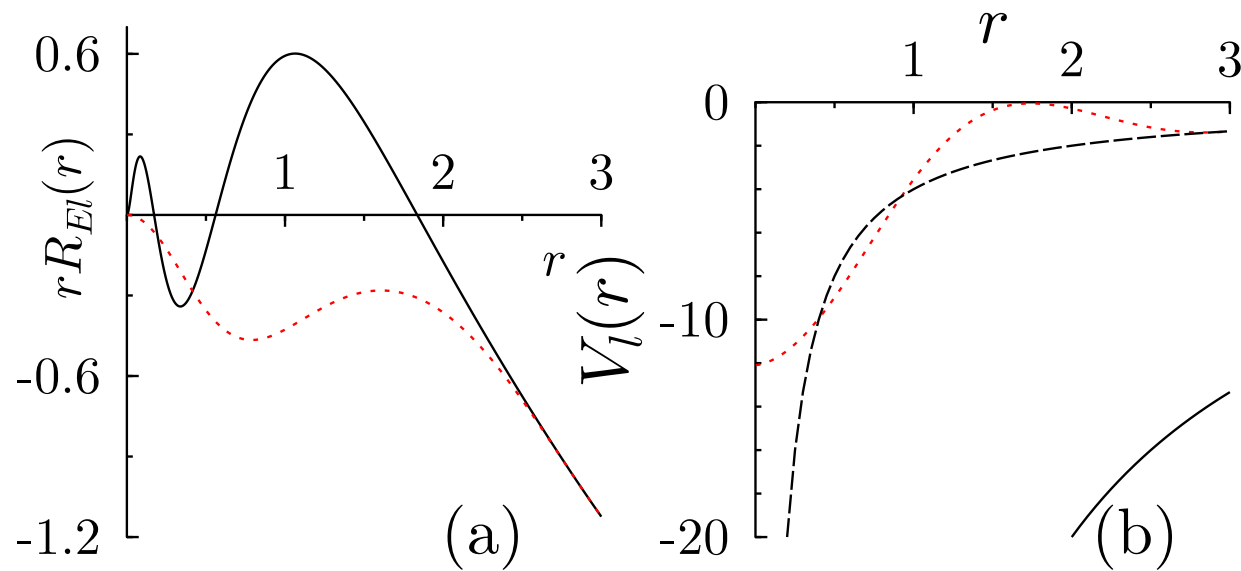
3. obtain pseudopotentials by inverted Schrödinger (Dirac) equation
4. “unscreen”: $V_l^{\text{ion}}(r) = V_l^{\text{scr}}(r) - V_{\text{H,XC}}[\rho^{\text{PS}}](r)$.

Ab-initio pseudopotential

- *Norm-conserving pseudopotentials*

Top and Hopfield [1973], Hamann et al [1979]

$$\left. \frac{\partial}{\partial \epsilon} \frac{\partial}{\partial r} \ln \psi_{\epsilon,l}(r) \right|_{\epsilon=E, r=R_C} = \frac{-2}{R_C^2 \psi_{E,l}^2(R_C)} \int_0^{R_C} |\psi_{E,l}(r)|^2 r^2 dr$$



norm-conserving PSP - continued

$$\int_0^{R_C} \left| \psi_{E,l}^{PS} \right|^2 r^2 dr = \int_0^{R_C} \left| \psi_{E,l}^{AE} \right|^2 r^2 dr$$



$$\left. \frac{\partial}{\partial \epsilon} \frac{\partial \ln \psi_{\epsilon,l}^{PS}(r)}{\partial r} \right|_{\epsilon=E, r=R_C} = \left. \frac{\partial}{\partial \epsilon} \frac{\partial \ln \psi_{\epsilon,l}^{AE}(r)}{\partial r} \right|_{\epsilon=E, r=R_C}$$

Equal amount of charge of ψ^{AE} and ψ^{PS} implies *equal logarithmic derivative, i.e. equal phase shift and equal scattering properties in the neighbourhood of E_{ref} to the 1-st order*

Ab-initio pseudopotentials - continued

- ... several various types of norm-conserving PSP
Generalized ... [Hamann 1989], [RRKJ 1990],
Extended ... [Shirley et al 1990]
- *Ultrasoft pseudopotentials*
Vanderbilt [1990]; relaxing norm-conserving condition
⇒ softer PSP, smaller R_C
- *PAW (Projector Augmented Wave) method*
Blöchl [1995]; (-)partial waves in the basis set, (-)still
frozen core, (+)unambiguous assignment between AE
and PS quantities

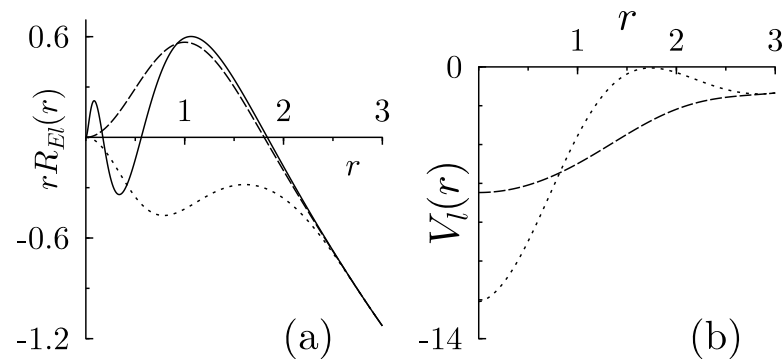
Desirable PSP properties ... and problems

1. “softness”

– means the size of basis (as small as possible) set that we need to achieve required accuracy in a given application;
is there any independent criterion ?

not discovered so far, but we have some “indicators”:

- Visual softness, curvature at $r = 0$



- Fourier image of PS-potential in reciprocal space
- Fourier image of PS-wave function in reciprocal space

2. “transferability”

intuitive meaning: the ability to work properly in different environments (solids, compounds, molecules),
e.g. Na in metal Na and NaCl

two main components:

- (a) the precision of reproducing the scattering properties of AE potential (log. derivative) as a function of ϵ in some neighborhood of E_{ref}
(energy transferability)
- (b) the precision of reproducing the AE eigenvalues under varying the external environmental conditions (i.e. charge density within DFT)

transferability ... - continued

(a) and (b) related via the *higher momenta* of the charge density [Shirley et al 1989] (related to higher energy derivatives of the phase-shift):

norm-conserving PSP

⇒ correct 1-st energy derivative of scattering properties

⇒ correct PSP behavior with respect to

making *first order* changes to external charge density

≡ *causing constant potential shift*

Sources of errors in pseudopotential method

- linearized method \Rightarrow higher order errors:
 - energy bands in a solid doesn't coincide with atomic eigenvalues
 - external charge density (caused by neighboring atoms) is not a *first order* change (doesn't cause constant potential shift)
 - \Rightarrow attempts to mimick the chemical bond by choosing *suitable* atomic configuration for generating pseudopotential
- next step in pseudopotential construction: subtracting the XC-potential of valence charge density: can cause **even first order error** since the XC-term is not linear; reduced by NLCC (non-linear core correction)

All electron pseudopotential (AEPP)

- selfconsistent pseudopotential-generating scheme that takes into account the solid state environment:
 1. all-electron atom reconstruction using the crystal boundary conditions
 2. constructing screened PSP: by minimizing a functional assembled from conditions to be satisfied
 3. unscreening the PSP by *real solid-state valence charge density*

AEPP - 1.atomic reconstruction

1. crystal charge density forms a *boundary condition*

$$\left. \frac{d}{dr} \ln \left(r^2 \left| \psi_{E_{\text{val},l,l}}^{\text{at}}(r) \right|^2 \right) \right|_{r=R_C} = \left. \frac{d}{dr} \ln \left(r^2 \rho_l^{\text{sps}}(r) \right) \right|_{r=R_C},$$

where partial charge density in a solid $\rho_l^{\text{sps}}(r)$ is evaluated by summing over all occupied states,

$$\rho_l^{\text{sps}}(r) = \sum_{\vec{k},n} \sum_{m=-l}^l \frac{1}{4\pi r^2} \int_{SPH} d\Omega d\Omega' \psi_{\vec{k},n}^*(r\hat{\mathbf{n}}) Y_{lm}(\hat{\mathbf{n}}) Y_{lm}^*(\hat{\mathbf{n}}') \psi_{\vec{k},n}(r\hat{\mathbf{n}}')$$

The boundary condition (above) replaces the standard condition for the wavefunctions to be normalizable ($\psi_{E,l}(r) \rightarrow 0$ for $r \rightarrow \infty$) and determines the eigenvalue E_l .

AEPP - 1. atomic reconstruction - continued

The normalization condition for the valence atomic-like radial wavefunctions is

$$\int_0^{R_C} \left| \psi_{E_{\text{val},l,l}}^{\text{at}}(r) \right|^2 r^2 dr = \int_0^{R_C} \rho_l^{\text{SPS}}(r) r^2 dr$$

(reversed norm-conserving condition)

The core states are recalculated selfconsistently with the valence (augmented to the crystal charge density) states.

AEPP - 2. screened PSP

2. each component $V_l^{\text{scr}}(r)$ satisfies

(i) At $r = R_C$ the potential $V_l^{\text{scr}}(r)$ matches the all-electron potential $V^{\text{at}}(r)$ up to the second derivative,

(ii) At $r = R_C$ the radial pseudo-wavefunctions $R_{E_{n,l},l}^{\text{ps}}(r)$ match the corresponding atomic-like radial functions by their values and first derivatives (for each energy window), and

(iii) the correct energy derivative of the pseudo-wavefunction is ensured by the norm-conserving condition (for main -valence- energy window; relaxed for semicore state)

AEPP - continued

differences with respect to conservative methods:

standard methods

AEPP

- *boundary conditions* for initial atomic calculation:
free, isolated atom derived from partial l -projected charge density in a solid
- *chemical bond* is taken into account in PSP
by intuitive, ad-hoc choice of atomic occupation numbers in natural way via the boundary condition
- *transferability ranges* are located
around the atomic eigenvalues around centers of l -projected DOS (bands) in a solid

- *core states*
atomic, frozen-core approximation self-consistent with the charge density in a solid
- *screening — XC potential term*
atomic valence charge density, NLCC self-consistent charge density in a solid