

The atomic, electronic, and magnetic structure of disordered metal surfaces - interpretation of the STM

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The research i am going to present on this occasion
has been accomplished
during my doctoral studies (2003-2008)
as a material for my PhD thesis

under the supervision of

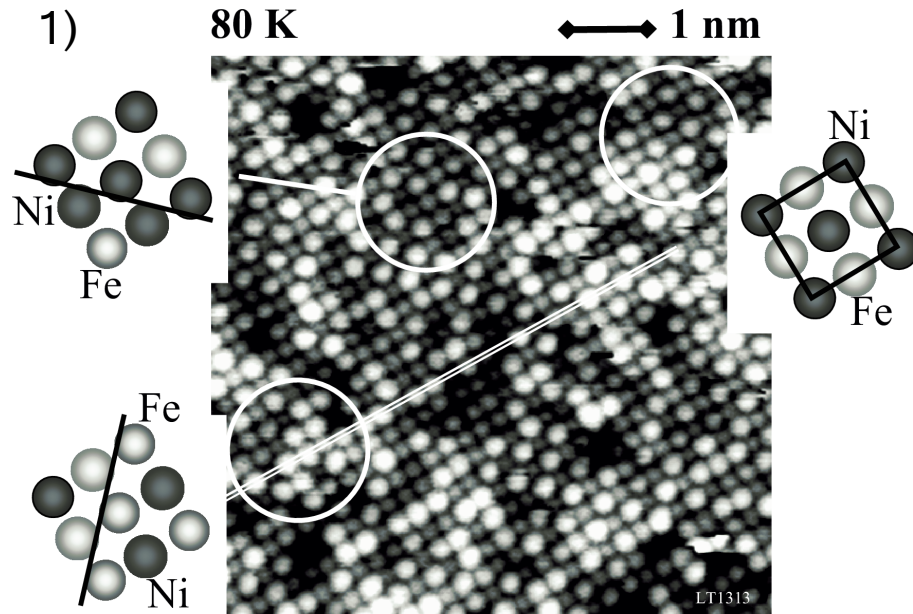
dr. František Máca

and in close collaboration with

dr. Josef Kudrnovský

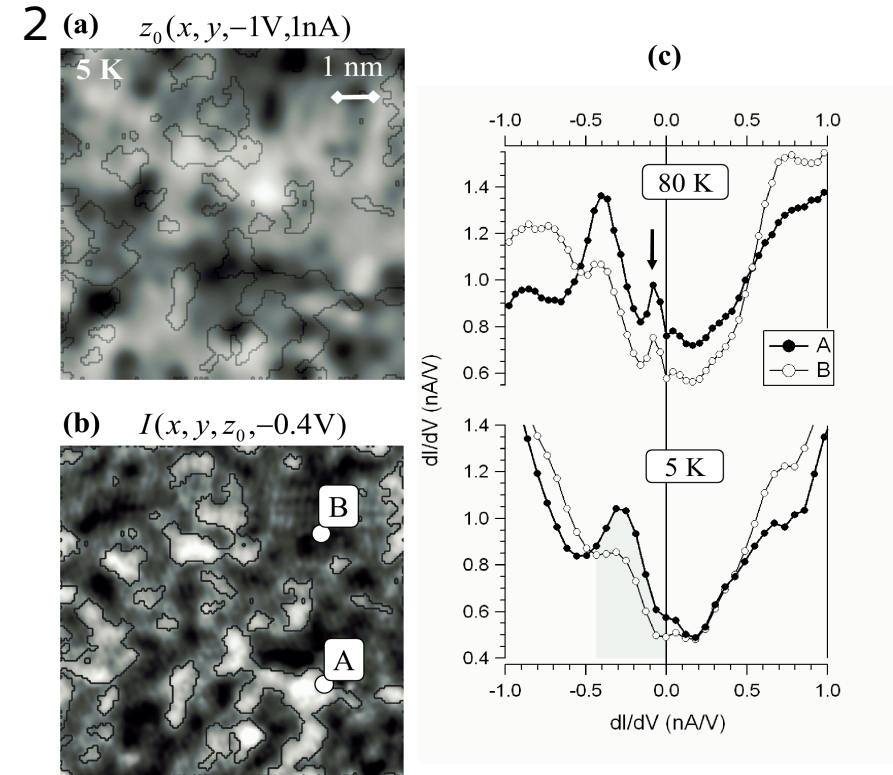
to both of whom I am largely indebted
along with all others not named here
who supported me during the time of my studies.

Experimental results for the Fe₆₄Ni₃₆ (100) surface



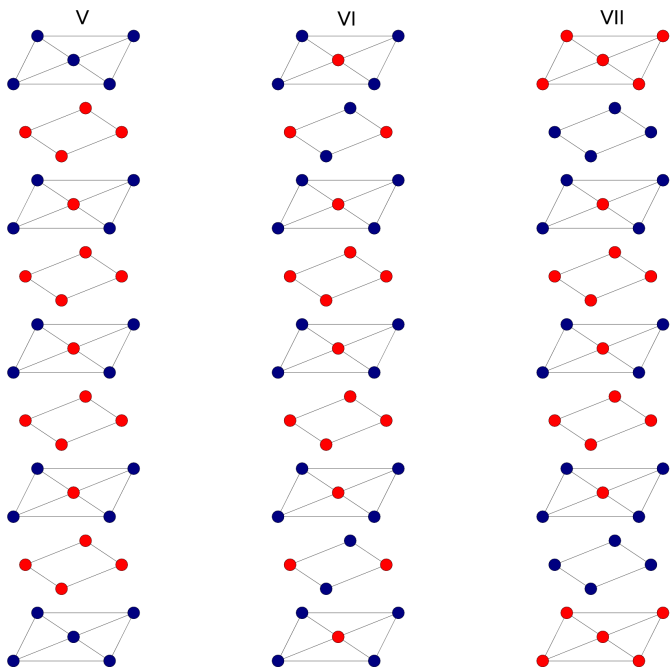
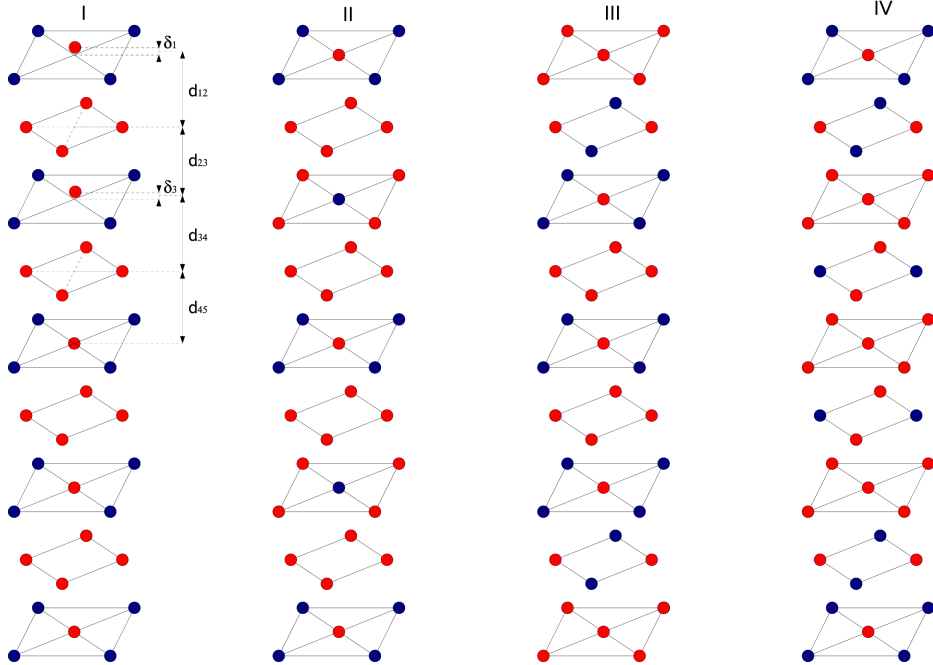
constant current STM image with atomic resolution, $I = 8 \text{ nA}$, $U = -2 \text{ mV}$

- 1) chemical resolution achieved (Fe brighter, i.e. “higher”, Ni darker, i.e. “lower”)
- 2) patches of the FeNi-c(2x2) structure prevail on the surface
- 3) peak of tunnelling conductance in STS at $U \approx -0.3 \text{ V}$



- (a) constant current STM image ($U = -1V$),
- (b) current map at -0.4 V (voltage corresponding to the STS peak)
- (c) typical STS curves (differential conductance vs voltage) averaged over selected areas on the surface

Models of the FeNi (001) atomic surface structure



FP-LAPW calculations (WIEN2k):

- atomic structure relaxation
- total energy calculation
- $d_0 = a/2 = 1.8 \text{ \AA}$

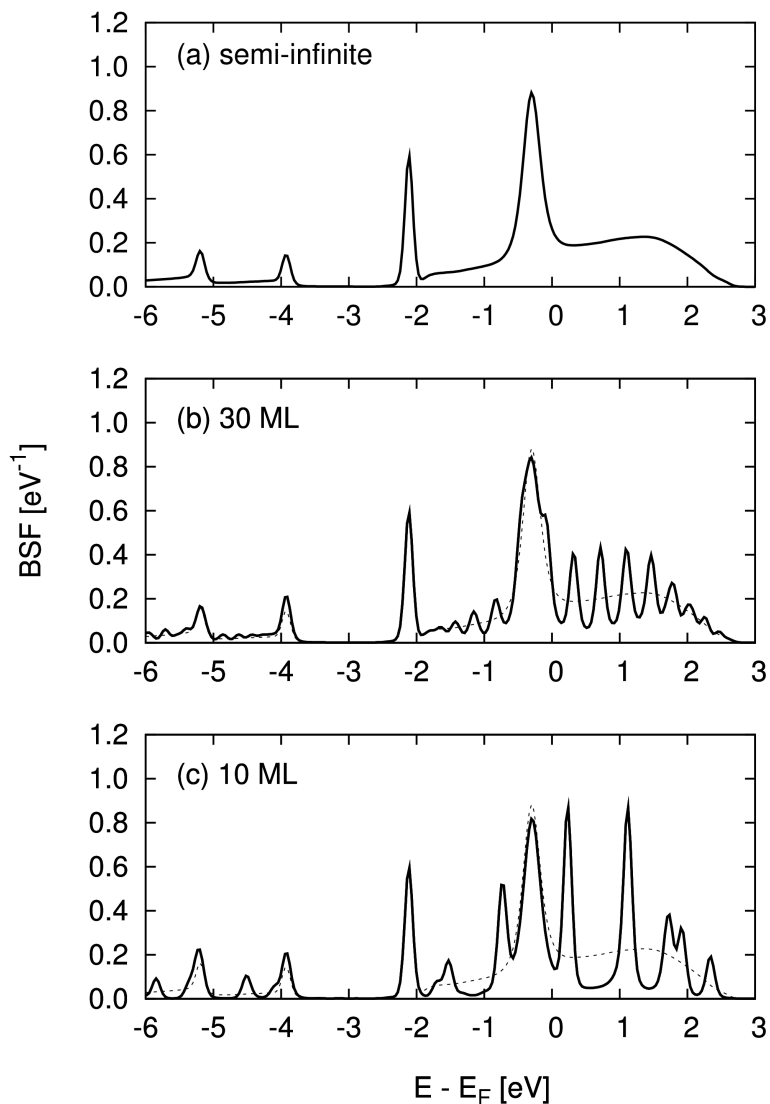
model #	I	II	III	IV
δ_1 / d_0	+3.3%	+7.5%	-4.1%	+2.8%
$(d_{12}-d_0) / d_0$	-1.0%	0.0%	+1.4%	-4.5%
δ_2 / d_0	0	0	+0.3%	+0.9%
$(d_{23}-d_0) / d_0$	1.4%	+2.6%	-2.3%	+2.6%
δ_3 / d_0	-1.0%	-0.1%	-0.4%	+1.5%
$(d_{34}-d_0) / d_0$	+1.2%	+1.9%	+2.3%	+1.5%
$E - E_I$ [meV]	0	+13	+39	-

model #	V	VI	VII
δ_1 / d_0	0.0%	+2.6%	+2.2%
$(d_{12}-d_0) / d_0$	-6.0%	-3.9%	-2.1%
δ_2 / d_0	0	+0.5%	0
$(d_{23}-d_0) / d_0$	+2.8%	-1.4%	-2.6%
δ_3 / d_0	-0.9%	+0.2%	+0.3%
$(d_{34}-d_0) / d_0$	+1.5%	+2.5%	+2.7%
$E - E_{VI}$ [meV]	+27	0	+127

- Fe atoms in the top surface layer always stick out, Ni atoms sit lower
- Deeper atomic layers are also influence the surface buckling.
- The $c(2 \times 2)$ -FeNi surface structure is especially favored in terms of total energy

The necessity of proper boundary conditions

The projected Bloch spectral function for $k_{\parallel} = 0$
1.8 Å above the Fe / Fe₆₄Ni₃₆ (100) surface
depending on the slab thickness

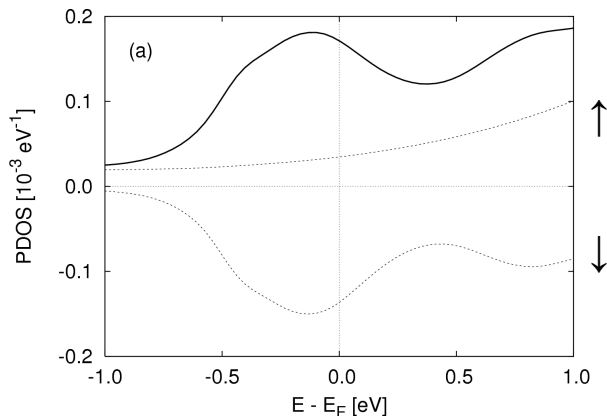


Method for the calculations that aim to interpret the STS:

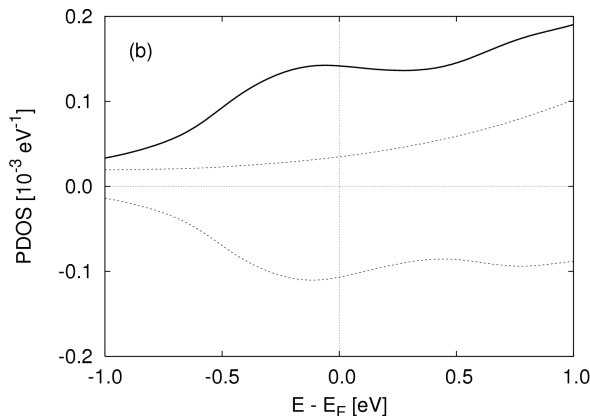
- TB-LMTO in ASA (tight-binding linear muffin-tin orbital method in atomic sphere approximation)
- Surface Green's function formalism (employed to account for the boundary conditions)
- CPA (coherent potential approximation) to treat the chemical disorder in the alloy

Projected densities of states (PDOS) and Bloch spectral functions for $k_{\parallel} = 0$ (BSF) at a site situated 5.4 Å above the surface (spin-resolved components and total)

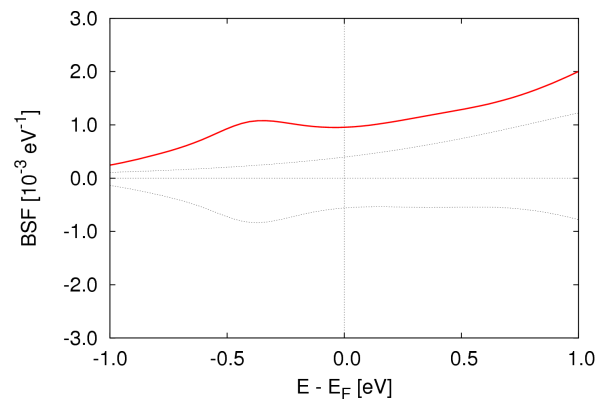
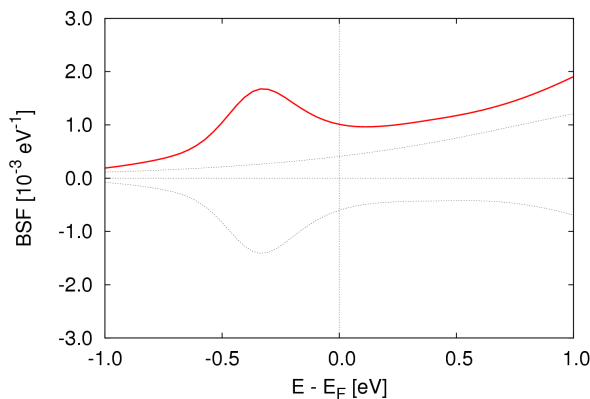
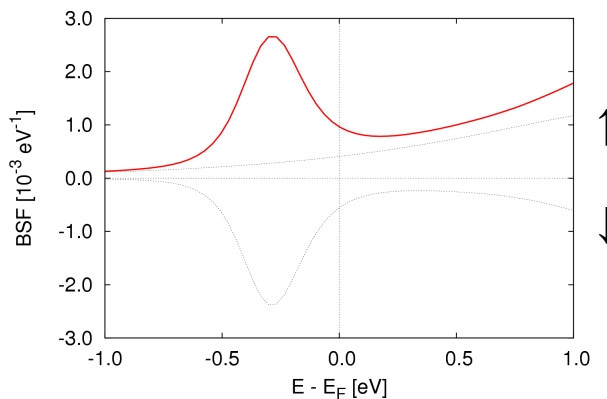
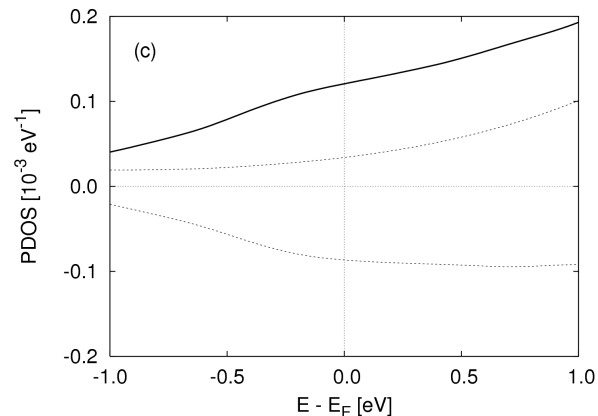
(a) Fe / Fe_{0.64}Ni_{0.36}



(b) Fe_{0.875}Ni_{0.125} / Fe_{0.64}Ni_{0.36}



(c) Fe_{0.75}Ni_{0.25} / Fe_{0.64}Ni_{0.36}



Spin components

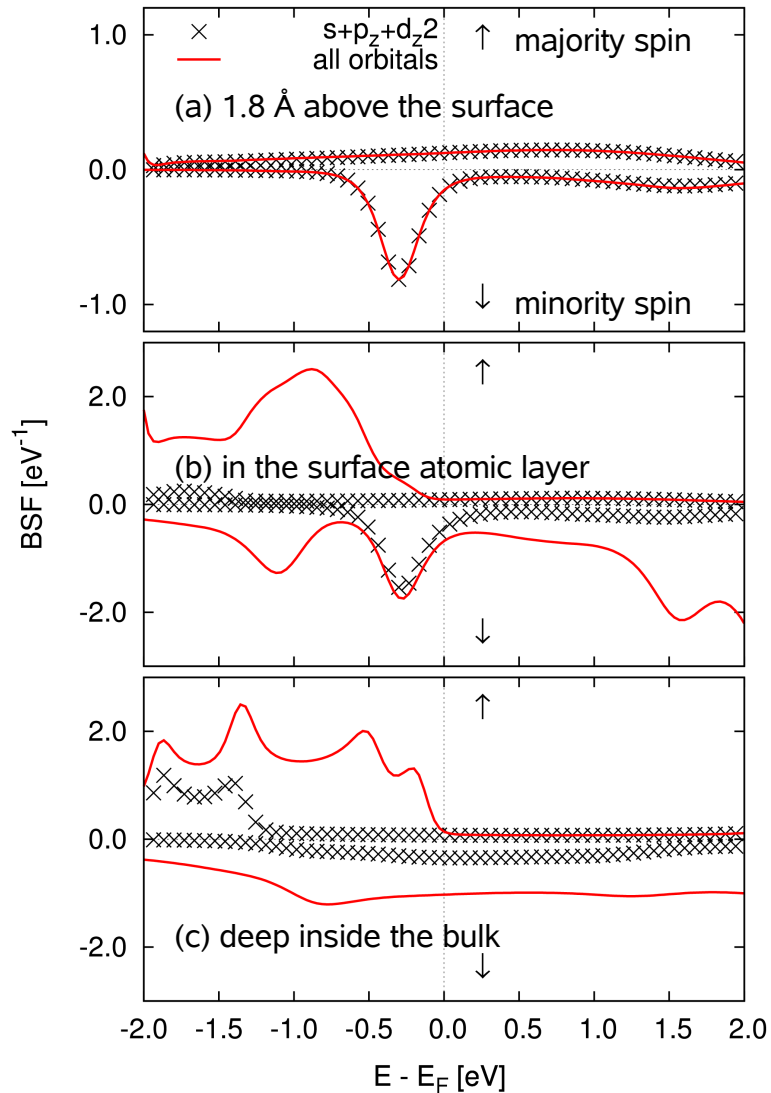
- positive half-plane: majority spin (↑)
- negative half-plane: minority spin (↓)
- full line: total (all spin)

There is a peak of the LDOS for the minority-spin electrons ≈ 0.3 eV below E_F originating mainly from the central part of the Brillouin zone, which appears in the vacuum above Fe-rich surfaces

Bloch spectral functions for $k_{\parallel} = 0$ (BSF) and projected densities of state (PDOS)

Fe / Fe_{0.64}Ni_{0.36}

Spin-resolved and orbital-projected BSF above the surface, on it, and beneath

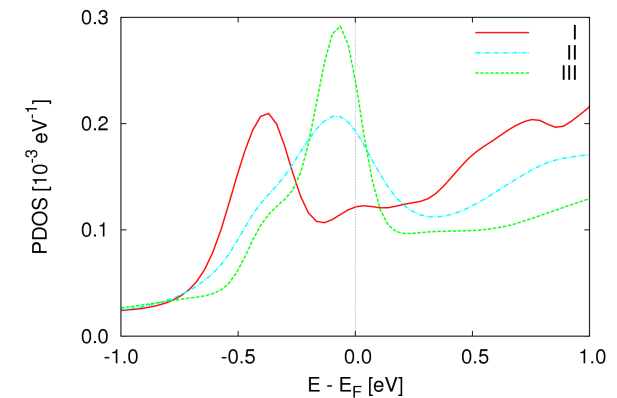
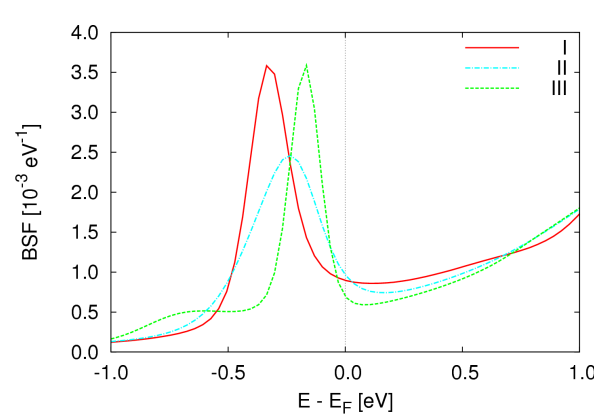


BSF for $k_{\parallel} = 0$ and PDOS 5.4 Å above the surface depending on the sub-surface composition

I. Fe / Fe / Fe_{0.64}Ni_{0.36}

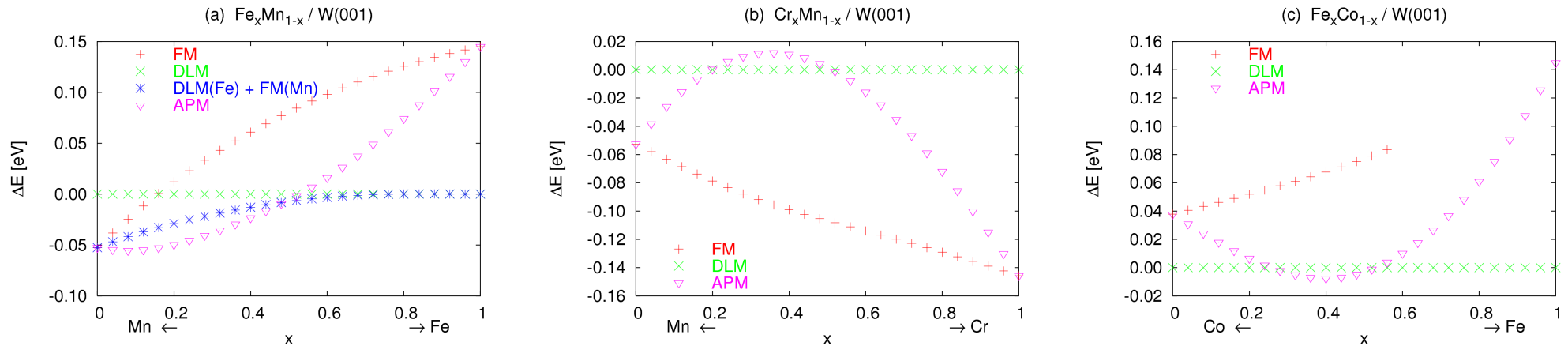
II. Fe / Fe_{0.5}Ni_{0.5} / Fe_{0.64}Ni_{0.36}

III. Fe / Ni / Fe_{0.64}Ni_{0.36}

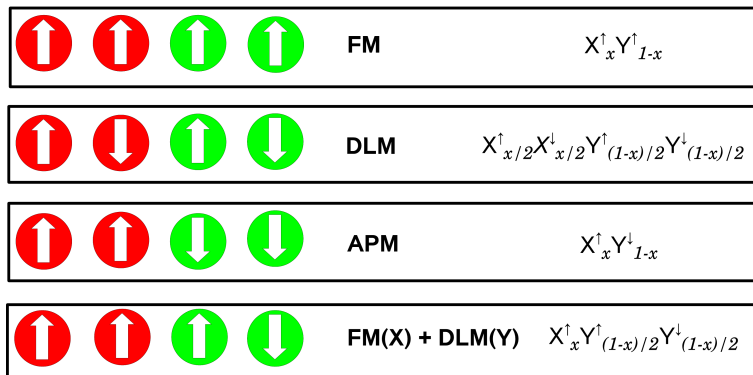


- The surface resonance of the minority-spin electrons is centered at $k_{\parallel} = 0$ and predominantly of a d_{z^2} -orbital character.
- Its existence depends mainly on the composition of the top surface atomic layer but its position on the energy scale also depends on the sub-surface layers.

Total energies of selected magnetic configurations for the binary surface alloys on W(100)



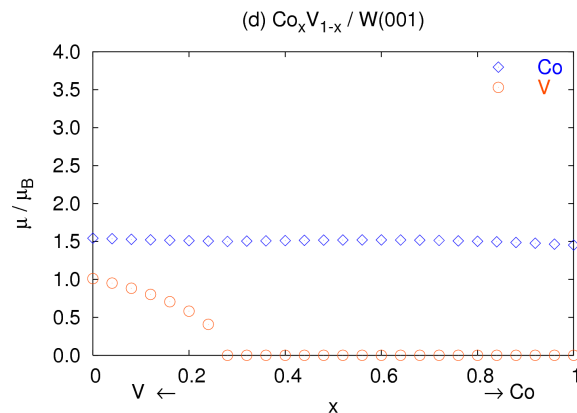
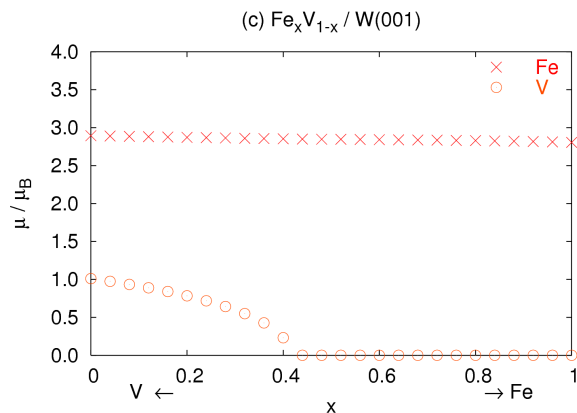
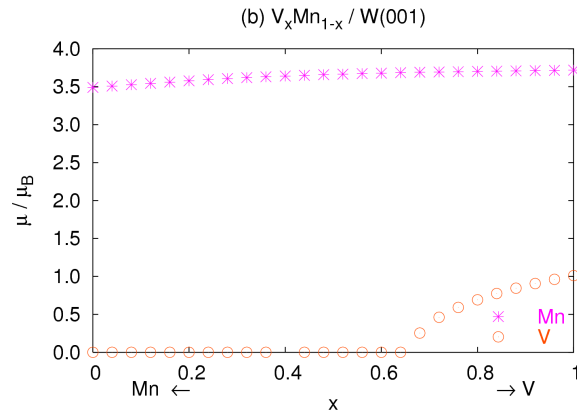
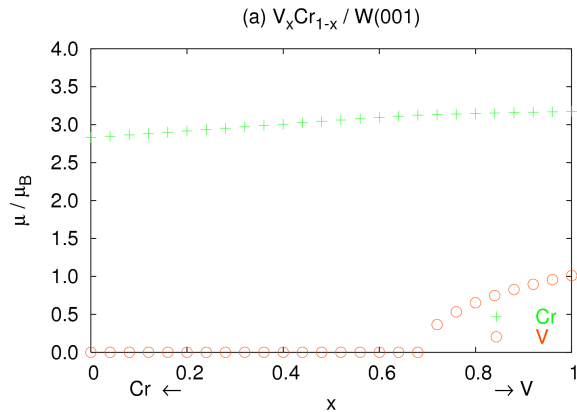
Selected magnetic configurations



Tendencies of alignment between magnetic moments of the 3d elements on the W(100) substrate

- parallel alignment for V, Cr, and Mn (including mixed pairs like (CrMn))
- anti-parallel alignment for Fe and Co (including FeCo)
- anti-parallel alignment also for FeMn, FeCr, CoMn, CoCr, and CoV
- weak tendency to parallel alignment also for FeV

Magnetic moments of atoms in the binary surface alloys



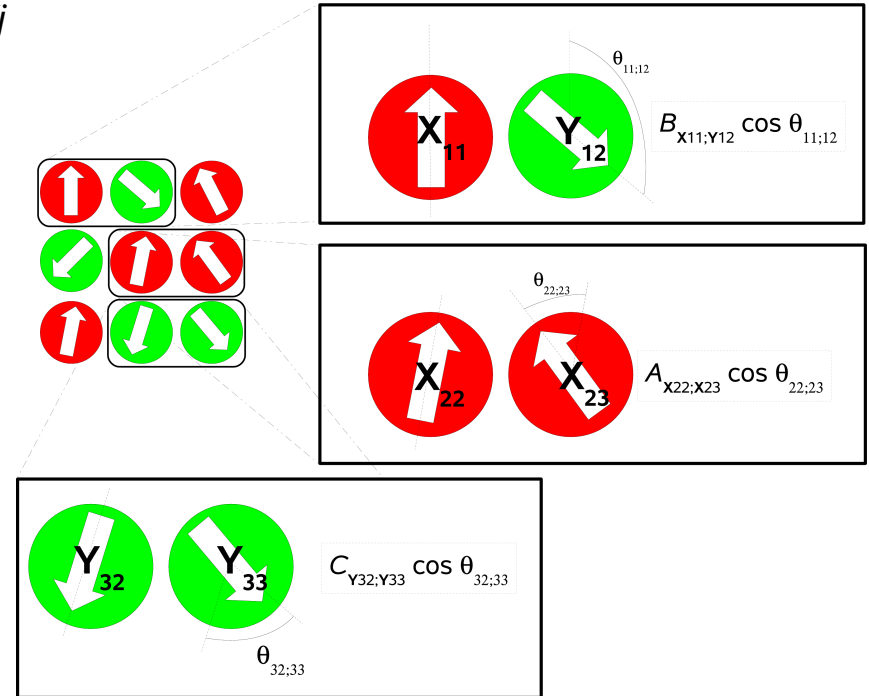
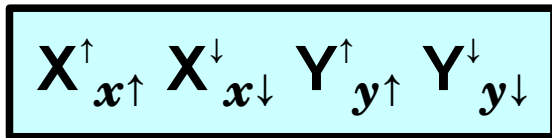
- *magnitudes of atomic magnetic moments of **Cr, Mn, Fe, and Co** change only **little** with magnetic configuration and chemical composition of the alloy (well localized moments)*
- *magnetic moments of **V** (vanadium) are very **sensitive** to both magnetic configuration and alloy composition (more itinerant character)*

Theoretical models for the magnetic interactions

$$E_{\text{mag}} = E_0 + \sum_{X_i, X_j} A_{ij} \cos \theta_{ij} + \sum_{X_i, Y_j} B_{ij} \cos \theta_{ij} + \sum_{Y_i, Y_j} C_{ij} \cos \theta_{ij}$$

We further restricted the models to

- **collinear** ($\theta_{ij} = \pm 1$) and
 - completely **disordered** (no correlation among sites),
- so they can be described as a four-component alloy with the **general formula**



Condition for thickness of 1 ML: $x^\uparrow + x^\downarrow + y^\uparrow + y^\downarrow = 1$

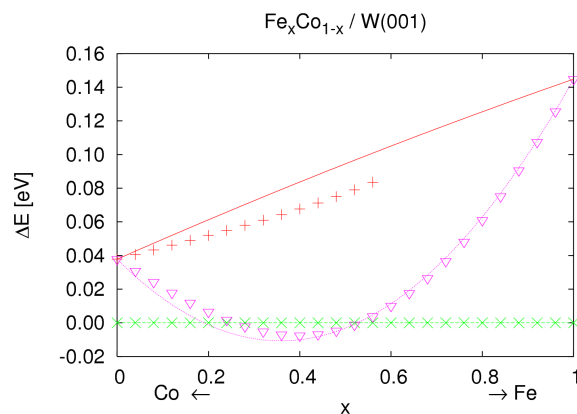
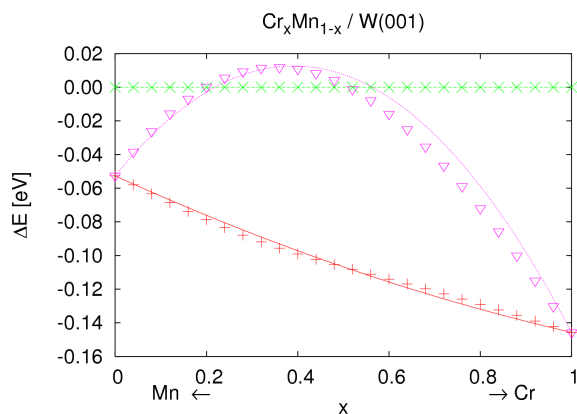
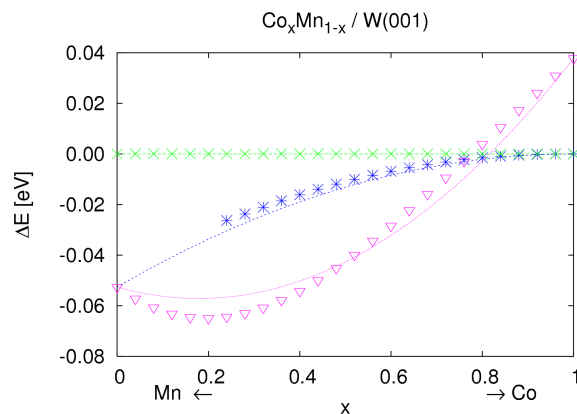
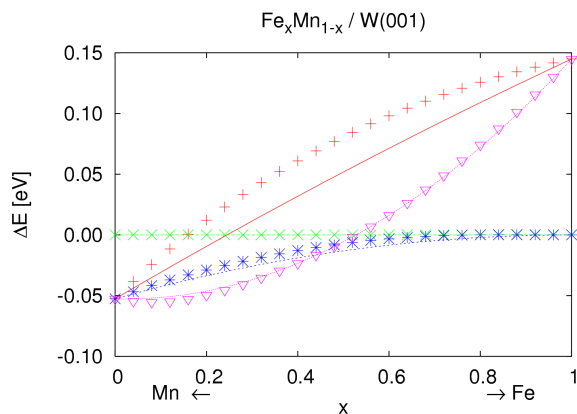
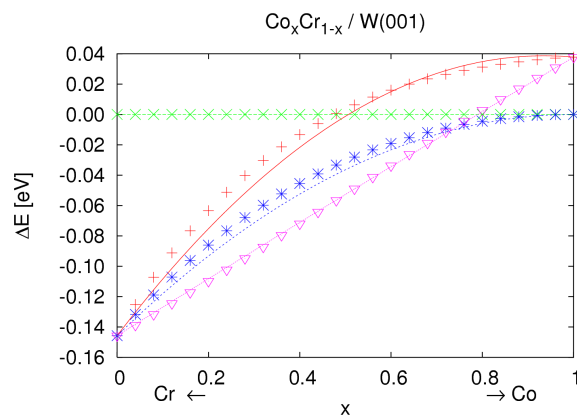
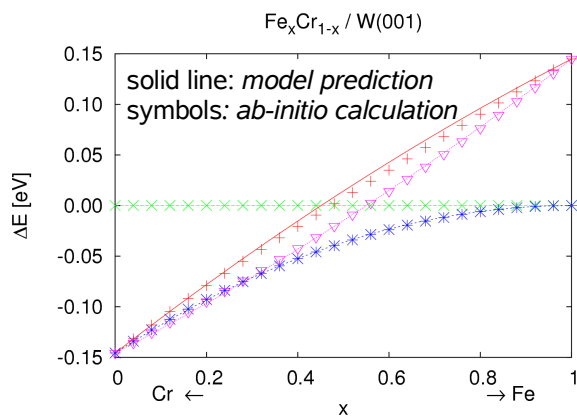
$$\bar{E} = \bar{E}_{\text{DLM}} + \frac{1}{2} J_{XX} (x^\uparrow - x^\downarrow)^2 + J_{XY} (x^\uparrow - x^\downarrow)(y^\uparrow - y^\downarrow) + \frac{1}{2} J_{YY} (y^\uparrow - y^\downarrow)^2$$

$$J_{XX} = \sum_{i \neq j} A_{ij} / N$$

$$J_{XY} = \sum_{i \neq j} B_{ij} / N$$

$$J_{YY} = \sum_{i \neq j} C_{ij} / N$$

Estimate of the model parameters and test of the model against *ab-initio* calculations



$$J_{\text{Cr-Cr}} = -0.292 \text{ eV}$$

$$J_{\text{Mn-Mn}} = -0.105 \text{ eV}$$

$$J_{\text{Fe-Fe}} = +0.290 \text{ eV}$$

$$J_{\text{Co-Co}} = +0.076 \text{ eV}$$

$$J_{\text{Cr-Mn}} = -0.229 \text{ eV}$$

$$J_{\text{Fe-Cr}} = +0.058 \text{ eV}$$

$$J_{\text{Fe-Mn}} = +0.115 \text{ eV}$$

$$J_{\text{Co-Cr}} = +0.103 \text{ eV}$$

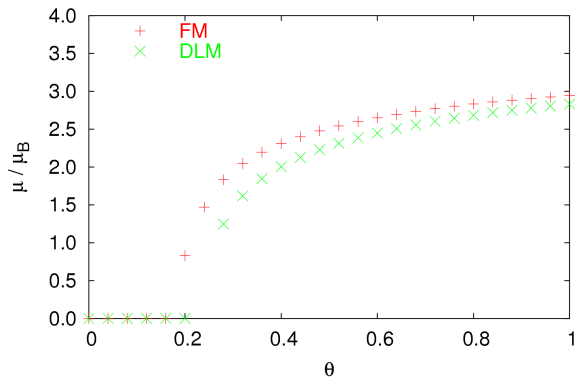
$$J_{\text{Co-Mn}} = +0.156 \text{ eV}$$

$$J_{\text{Fe-Co}} = +0.195 \text{ eV}$$

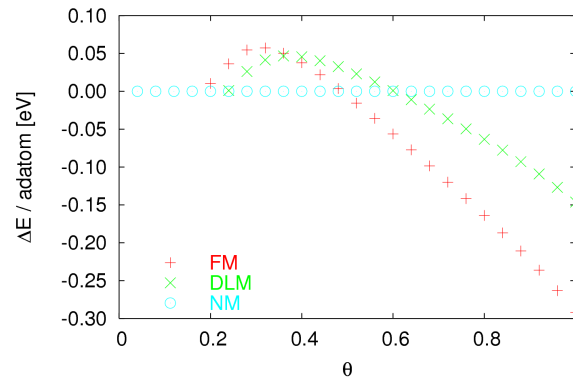
Estimates of the configuration with the lowest total energy (among those of the $X_{x\uparrow}^\uparrow X_{x\downarrow}^\downarrow Y_{y\uparrow}^\uparrow Y_{y\downarrow}^\downarrow$ type) also well agree with *ab-initio* results

Magnetic moments (μ) at submonolayer coverage (θ): Cr / W(100)

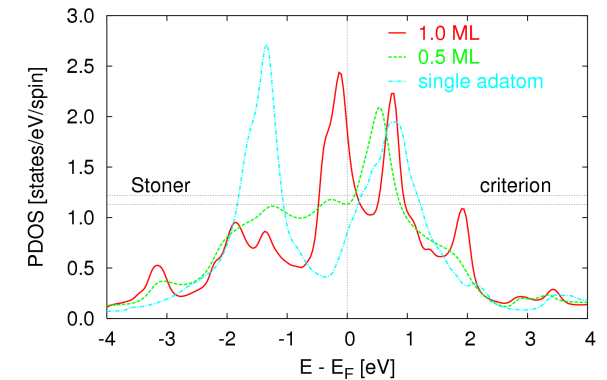
magnetic moments
in magnetic states



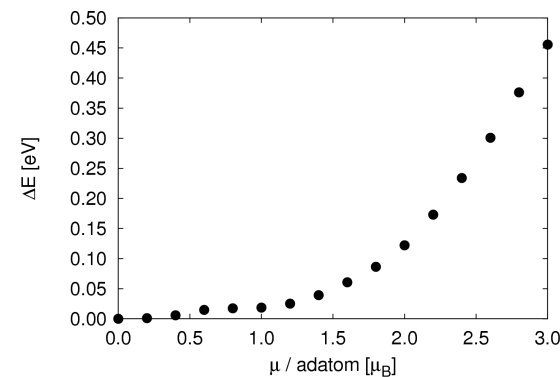
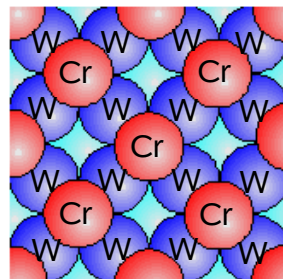
total energies of various
magnetic configurations



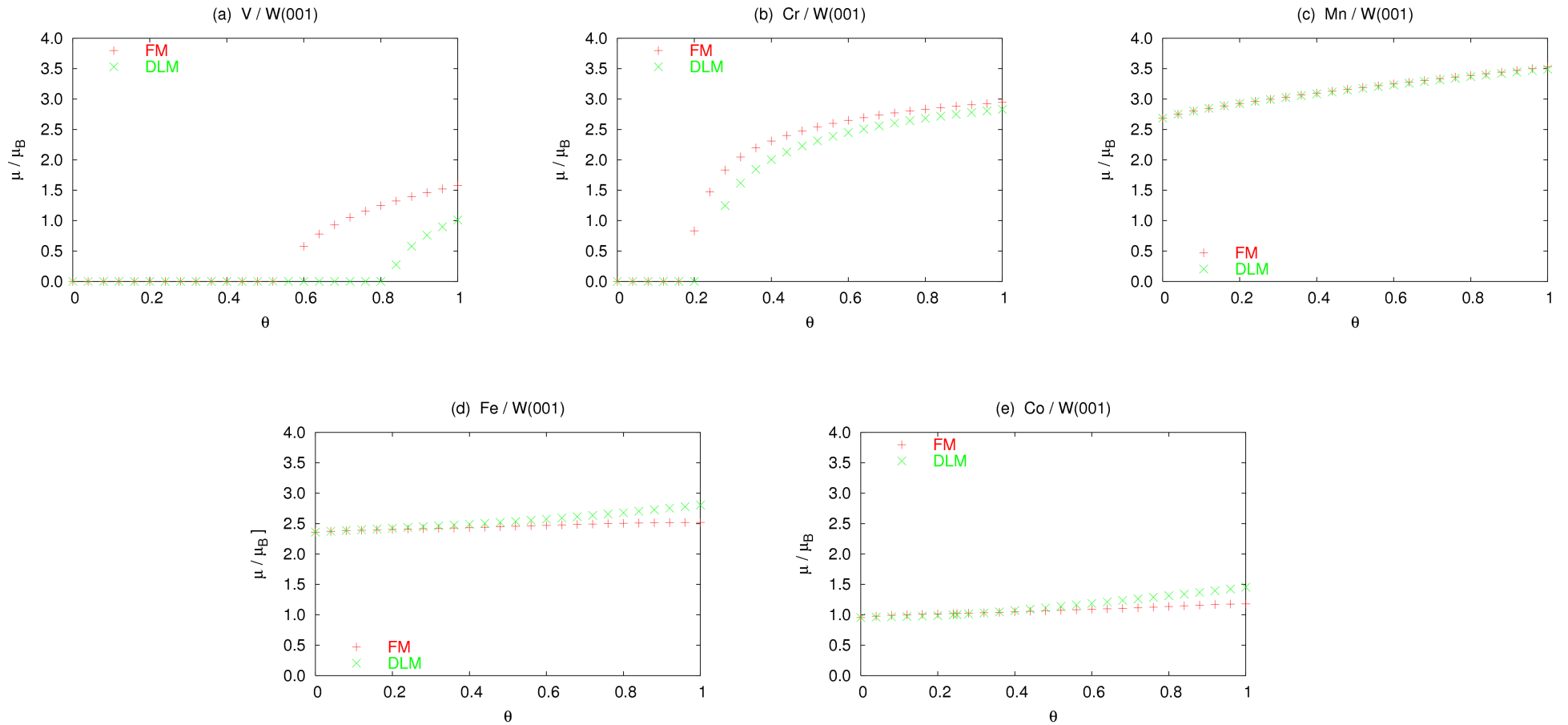
Stoner criterion for Cr / W(100):
 $I \cdot \text{DOS}(E_F) > 1$



Fixed-spin FP-LAPW calculations for the $c(2 \times 2)$ -Cr / W(100) structure:
verification of the non-magnetic ground state



Magnetic moments (μ) at submonolayer coverage (θ): overview for all elements



Surface magneto-crystalline anisotropy

Preferential orientation of the magnetic axis has been studied for monolayers of V, Cr, and Mn on W(100) (and for the Fe / W(100) monolayer previously by Kubetzka *et al.*).

- Spin-orbit coupling and dipole-dipole interaction has to be included.
- Surface Brillouin zone sampled by a mesh of 60x60 k -points.
- Plane waves up to $E_{\text{kin}} = 17$ Ry in the basis set.

composition of the surface layer	MAE per surface atom (MAE = $E_{\parallel} - E_{\perp}$)
V / W(100)	-2.0 meV \parallel
Cr / W(100)	+2.6 meV \perp
Mn / W(100)	+5.1 meV \perp
Fe / W(100)	+2.4 meV \perp

- Dipole-dipole interaction is almost negligible.
- *Shick et al.*: The large anisotropy of Mn/W(100) is caused by
 1. large SO-coupling at W atoms.
 2. strong exchange coupling between the Mn overlayer and the top W layer.
 3. large magnetic moment of Mn atoms.

Conclusion

The utility of the *ab-initio* calculations as an aid to the analysis of surface structures by the STM has been demonstrated for:

1) Invar

- The observed apparent corrugation in the STM images (chemically sensitive contrast between atoms as well as variations on a large scale) has been related to real surface buckling (actual shift of the atomic positions, Fe up and Ni down).
- The peak in the STS spectra was explained for Fe-enriched areas on the surface by a surface resonance at energy ≈ 0.3 eV below the Fermi level, arising near the center of the Brillouin zone, and derived predominantly from the d_{z^2} orbitals.

2) Magnetic overlayers of 3d metals on W(100)

- Magnetic moments of Cr, Mn, Fe, and Co are almost constant. Surface alloys of these elements may be described along the lines of the Heisenberg or Ising model.
- Magnetic moment of V is very sensitive to its environment.
- Surface magnetocrystalline anisotropy: in-plane alignment for V, normal-to-plane for Cr, Mn, and Fe. Especially large anisotropy found for Mn/W(100).
- The interplay of competing alignment tendencies (different signs of J_{XY} and of the magnetic anisotropy) may lead to complex non-collinear magnetic structures.